

# ハーフメタルのスピン트로ニクスデバイス応用と

## 軟X線MCD解析

*- Applications of half-metals to spintronics devices and  
utilizations of soft X-ray magnetic circular dichroism -*

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- **Introduction** : *Half-metallic materials*  
*Half-metallic Heusler alloys*

- **Experiments**

*Experiment 1*  $\text{Co}_2\text{MnSi}$ -based magnetic tunnel junctions

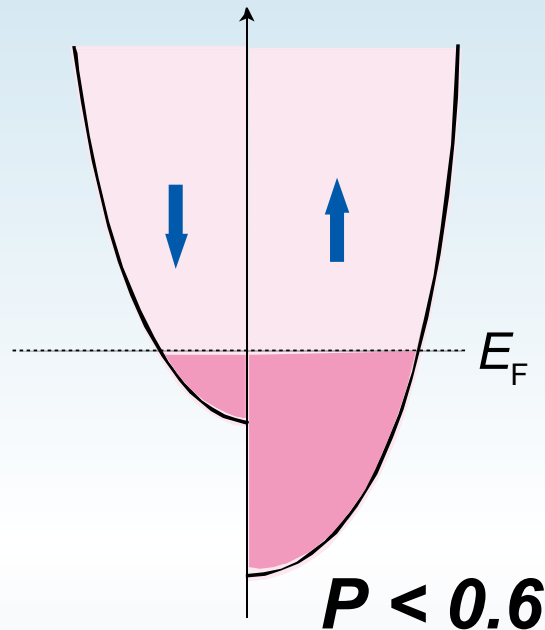
*Experiment 2* MTJs with  $L1_0/L2_1$ -ordered alloy hybrid electrodes

*Experiment 3* Half-metallic Ferri-magnets :  $\text{Mn}_2\text{VAI}$

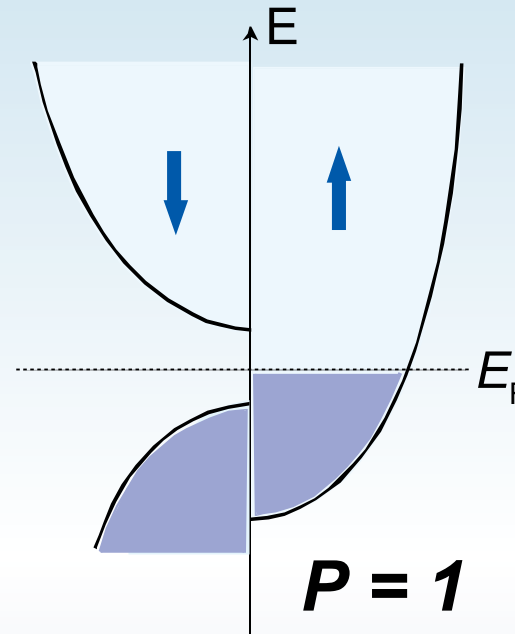
*Experiment 4*  $\text{Co}_2\text{MnSi}$ -based CPP-GMR devices

- **Summary**

## 3d Ferromagnet



## Half-metal



Spin-polarization

$$P = \frac{D_{\uparrow}(E_F) - D_{\downarrow}(E_F)}{D_{\uparrow}(E_F) + D_{\downarrow}(E_F)}$$

Candidates of half-metallic materials

Oxides :  $\text{CrO}_2$ ,  $\text{Fe}_3\text{O}_4$ ,  $\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3$  etc.

Zinc-blend type compounds:  $\text{CrAs}$ ,  $\text{MnAs}$  etc.

Half-Heusler alloys:  $\text{NiMnSn}$ ,  $\text{PtMnSb}$  etc.

Full-Heusler alloys :  $\text{Co}_2\text{MnSi}$ ,  $\text{Co}_2\text{MnGe}$ , etc.

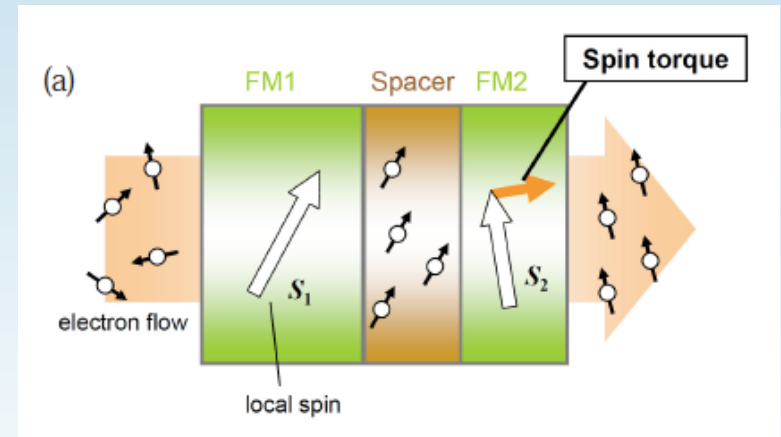
• MTJs & CPP-GMR devices

$$TMR \text{ ratio} = \frac{2P_1P_2}{1 - P_1P_2} \times 100 (\%)$$

Spin-torque induced magnetization switching

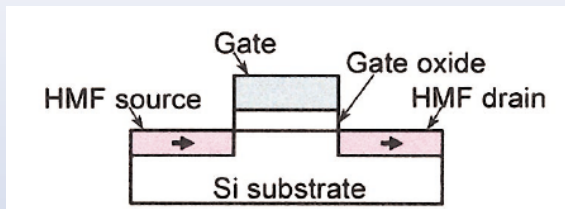
$$J_C \propto \frac{\alpha}{g(\theta)} M_s^2 t$$

Spin-injection efficiency :  $g(\theta) = \frac{P_1}{1 + P_1P_2 \cos \theta}$

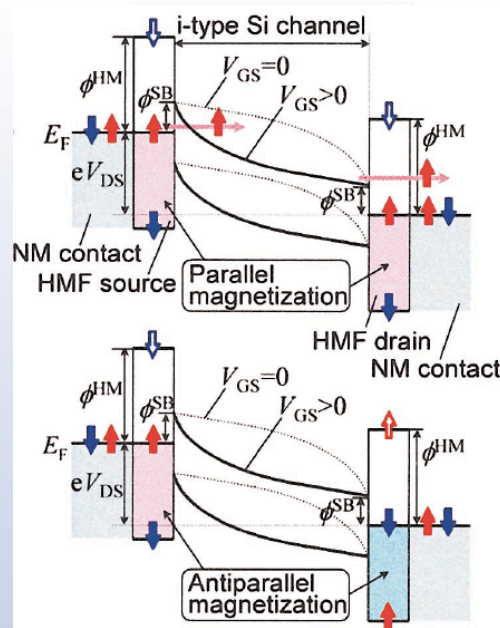


• Spin-Transistor

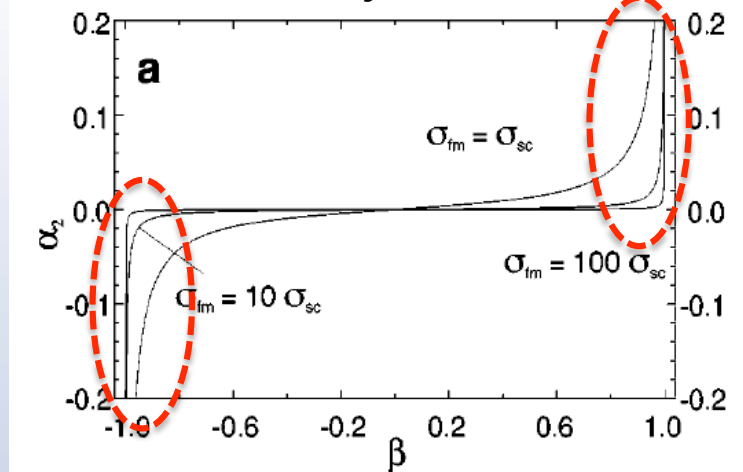
Spin-MOSFET based on Half-metallic source and drain



Sugahara et al. APL (2004)



FM/SC/FM junction

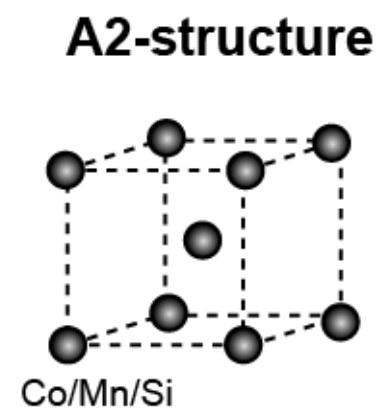
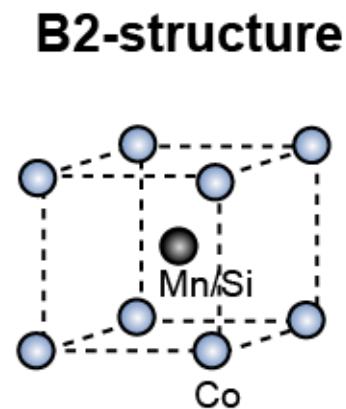
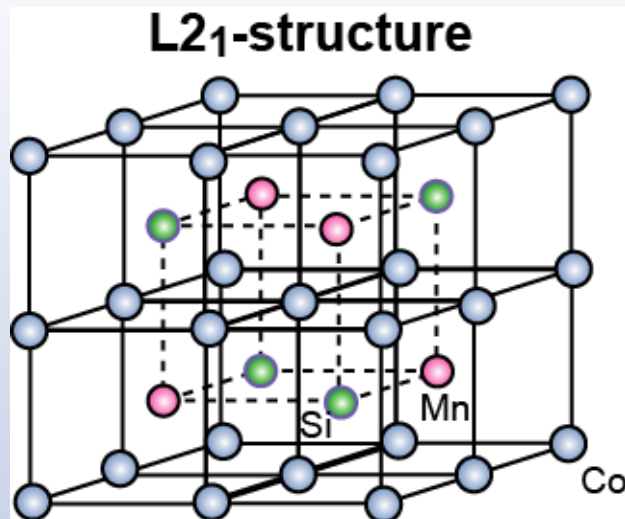
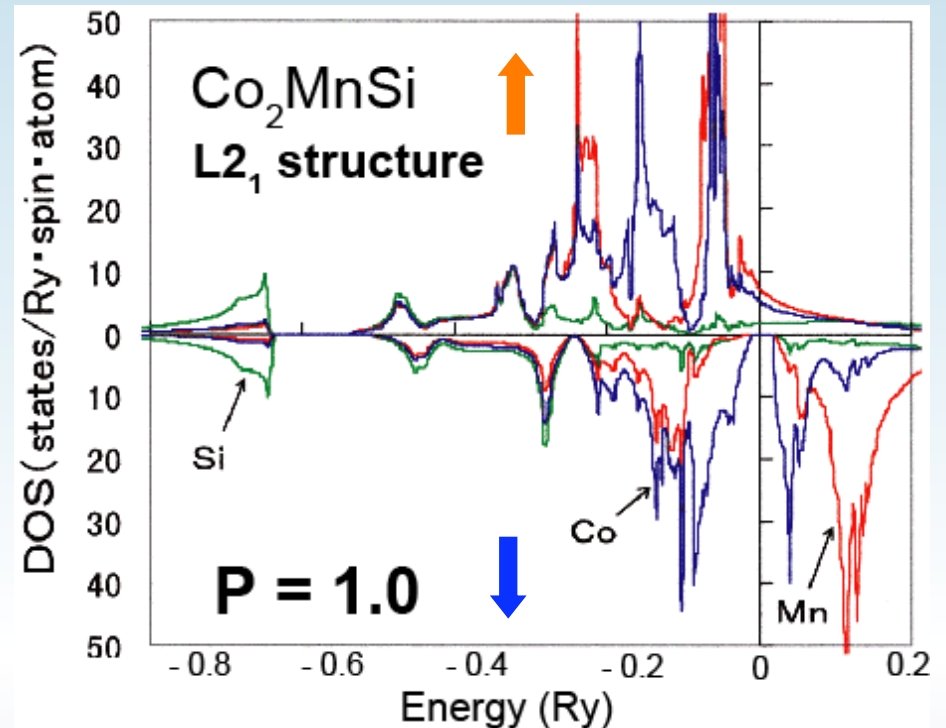


Schmidt et al. PRB (2000)

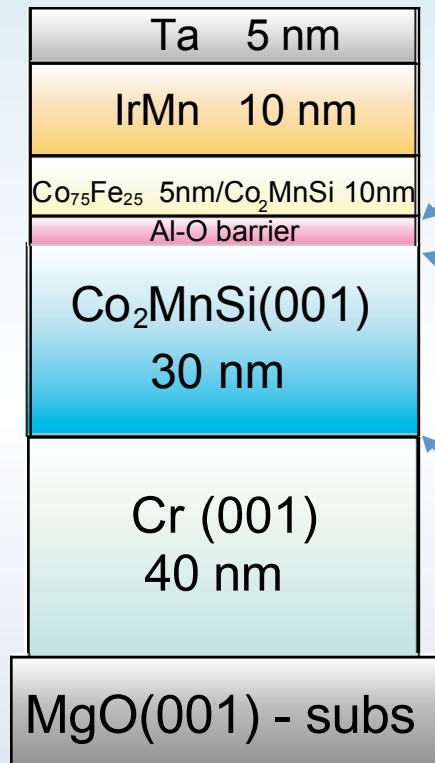
## $\text{Co}_2\text{MnSi}$ (CMS)

- High Curie temp.:  $T_C \sim 985$  K
- Half-metallic energy gap  $E_G$ :  $0.4 \sim 0.6$  eV
- High degree of  $L2_1$  chemical ordering is easily obtained

*CMS is a promising candidate in the half-metal candidates.*



UHV magnetron sputtering system ( $P < 1 \times 10^{-7} \text{Pa}$ )



**MTJ**

Al-O barrier : Plasma oxidation of Al layer  
 Micro-fabrication : PL & Ar ion milling  
 (element size :  $5 \times 5 - 100 \times 100 \mu\text{m}^2$ )  
 Annealing :  $200 - 500^\circ\text{C}$  with applying  $H$   
 MR, I-V curve : DC 4-probe method

**Co<sub>2</sub>MnSi bottom epitaxial electrode**

Composition : Target  $\text{Co}_{2.00}\text{Mn}_{1.28}\text{Si}_{1.30} \rightarrow$  Film :  $\text{Co}_{2.00}\text{Mn}_{1.00}\text{Si}_{1.08}$   
 Deposited@RT  $\rightarrow$  Post-annealed@ $T_a = 250 - 650^\circ\text{C}$   
 Crystal structure, the degree of ordering : XRD  
 Magnetization : SQUID, VSM  
 Surface image : AFM

**Cr buffer layer**

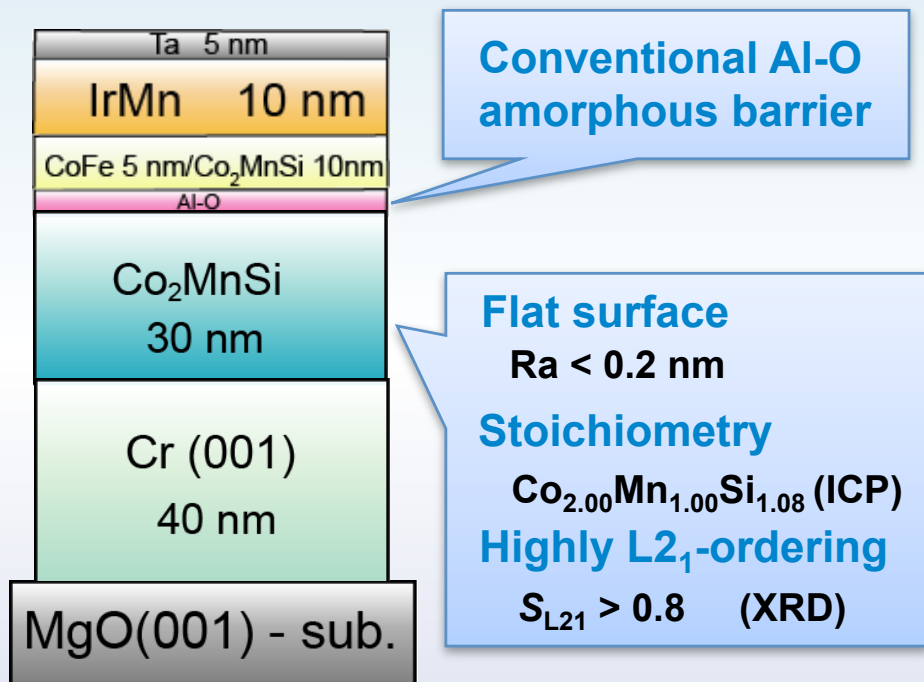
Deposited@RT  $\rightarrow$  Post-annealed@ $700^\circ\text{C}$

Epitaxial relation :

$\text{MgO}(001)\langle 100 \rangle / \text{Cr}(001)\langle 110 \rangle / \text{CMS}(001)\langle 110 \rangle$

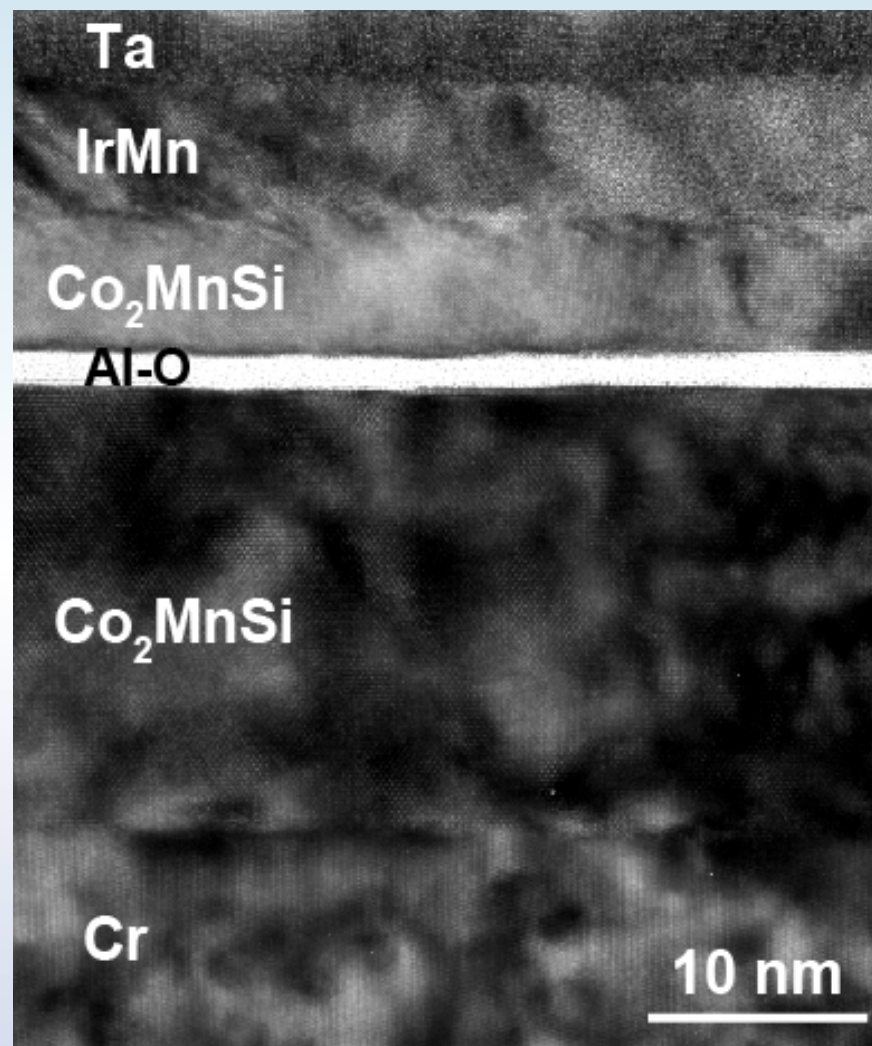
Lattice misfit < 5%

Magnetic tunnel junction (MTJs) with epitaxial CMS bottom electrode



High-quality epitaxial CMS electrode

HR-TEM image



*Extremely flat and sharp interfacial morphology*

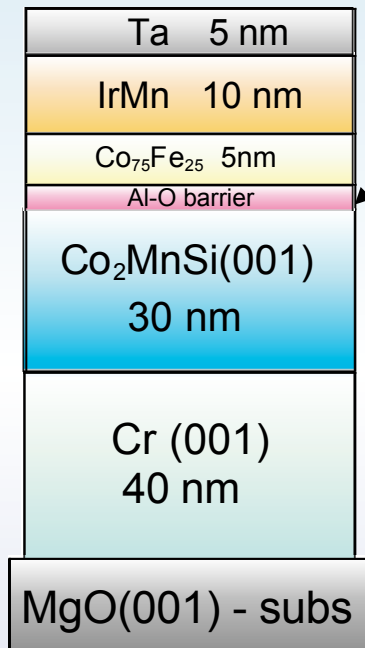
Co<sub>2</sub>MnSi : Mn has large affinity to oxygen compared to conventional 3d FM.



Magnetic impurities easily create between CMS/Al-O interface.



Optimization of interfacial chemical conditions is necessary.



1. Plasma oxidation time for Al-O :  $t_{\text{ox}} = 10 - 240 \text{ s}$

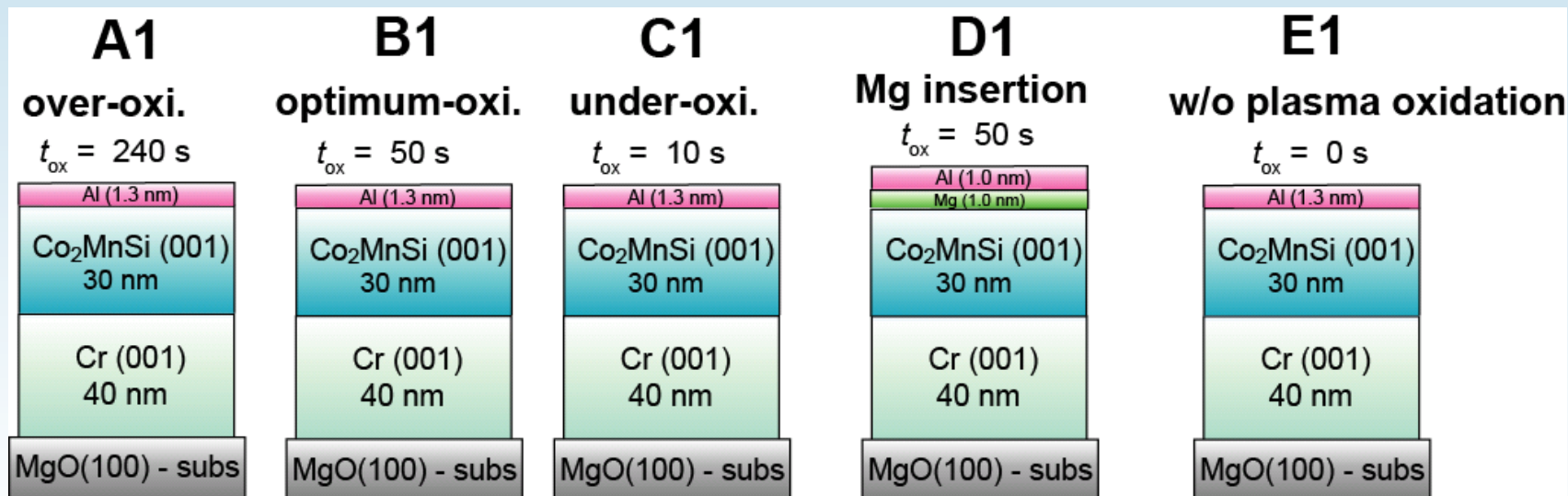
2. Insert thin Mg layer between CMS and Al-O.

(Formation energy of oxide : Mg < Al, Co, Mn, Si)

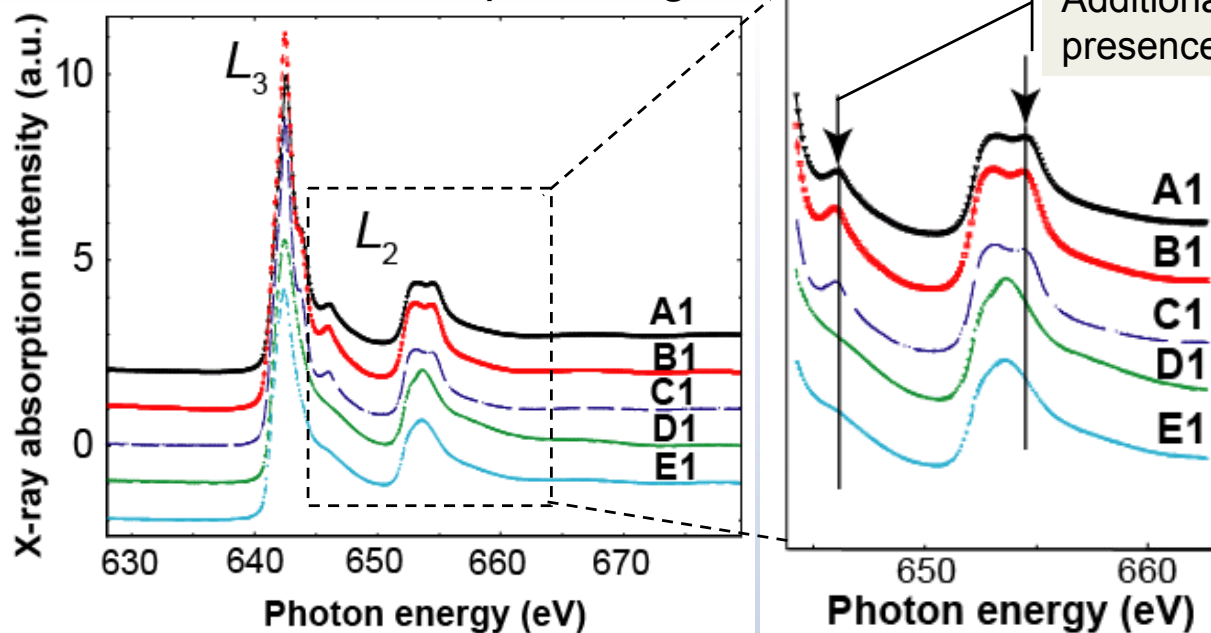
Thickenss of Mg layer :  $t_{\text{Mg}} = 0 - 1.3 \text{ nm}$



Investigation of interfacial Mn impurities by XAS using TEY (ALS beamline 6.3.1)



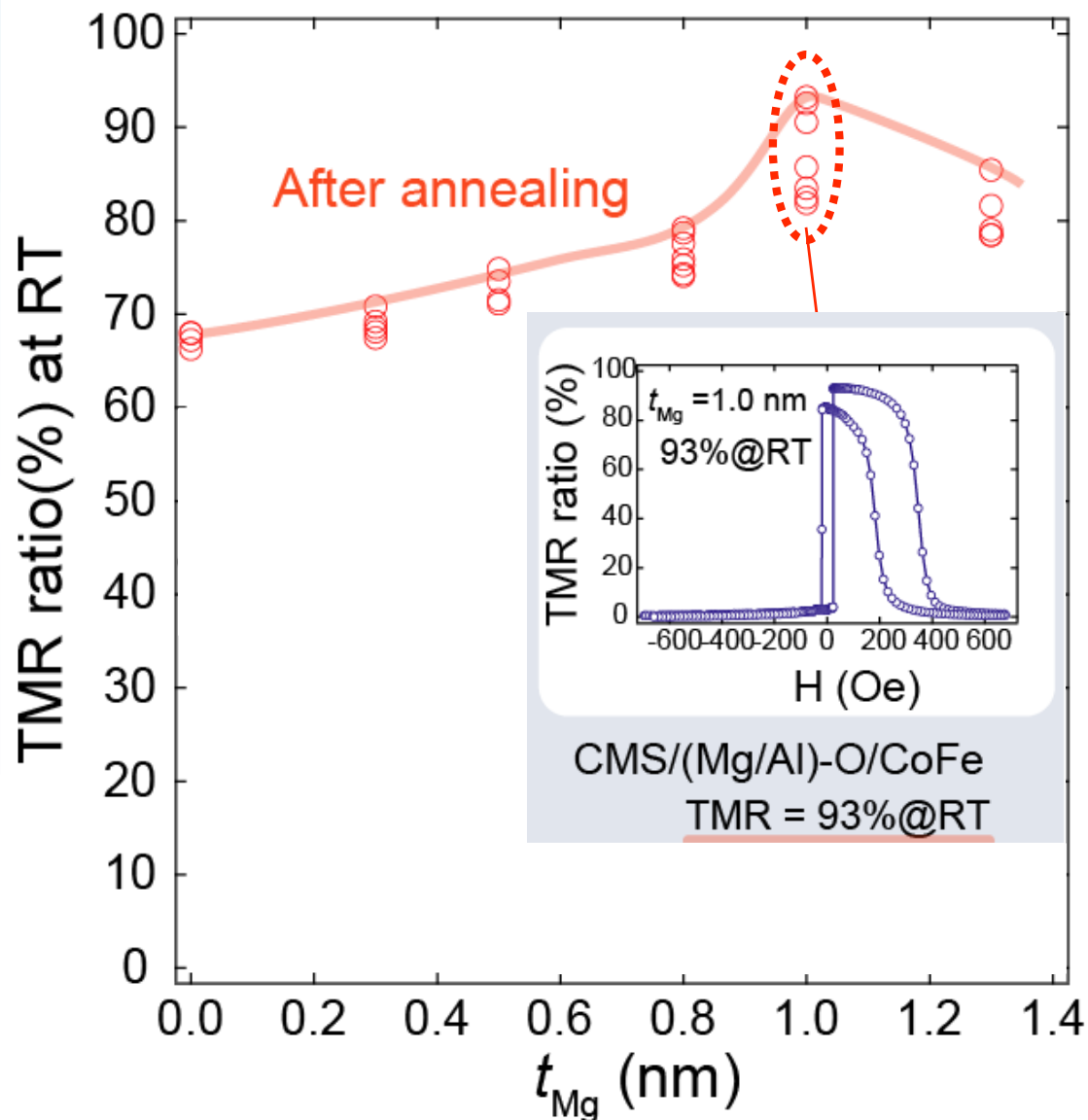
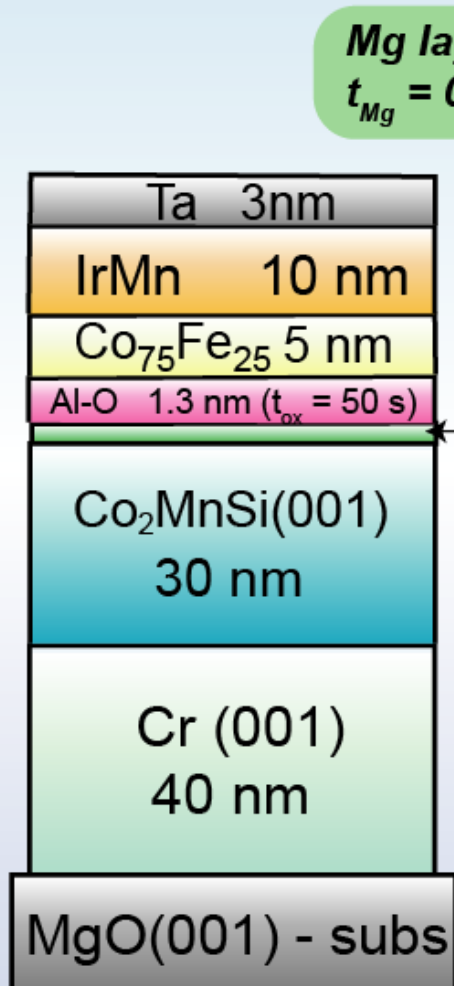
XAS :Mn L-absorption edges



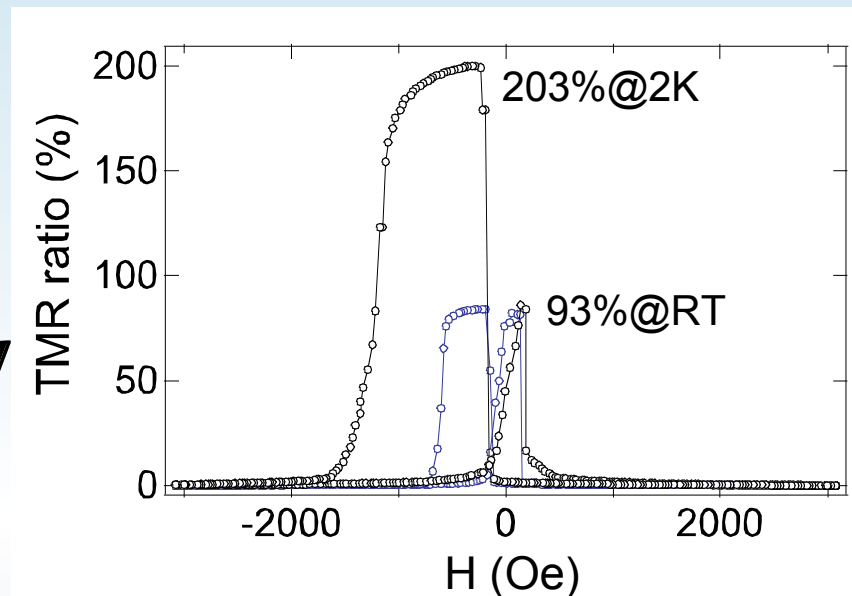
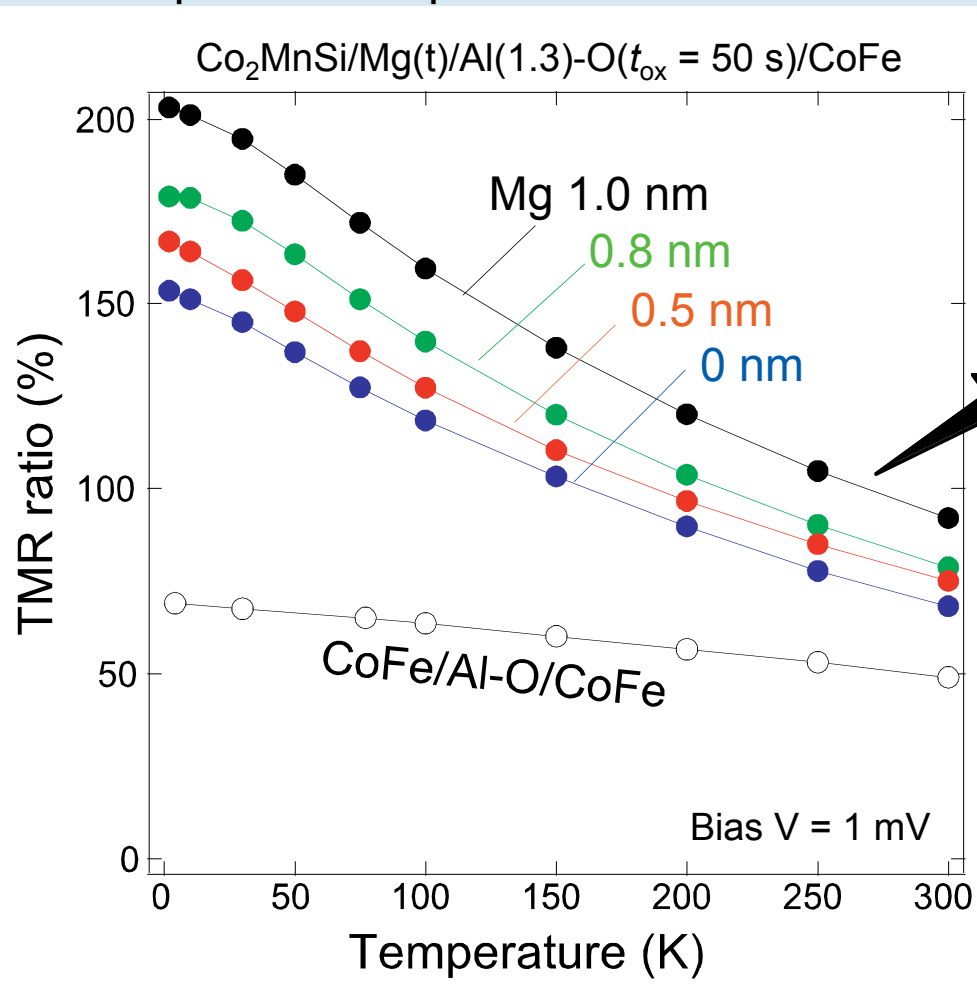
**A1, B1, C1:**  
large amount of Mn-O at interface

**D1, E1:**  
few Mn-O at interface

Mg thickness dependence of TMR ratio



Temperature dependence of TMR ratio



CMS/Mg/Al-O/CoFe  
TMR = 203% (T = 2K)



$$0.50 < P_{\text{CoFe}} < 0.52$$

$$0.97 < P_{\text{Co2MnSi}} < 1.00$$

**Nearly ideal half-metallic spin-polarization**

Sakuraba et al. J. Mag. Soc. Jpn. (2007)

**XAS using TEY is an useful technique to investigate barrier interface quality in MTJs**

### $\text{Co}_2\text{MnSi}$ based-MTJs

#### $\text{Co}_2\text{MnSi}/\text{Al-O}/\text{Co}_2\text{MnSi}$ -MTJ

TMR = 67%(RT), 570%(2K)

Sakuraba *et al.* *APL* 88, 192508 (2006)

#### $\text{Co}_2\text{MnSi}/\text{MgO}/\text{CoFe}$ -MTJ

TMR = 217%(RT), 753%(10K)

Tsunegi *et al.* *APL* 93, 112506 (2008)

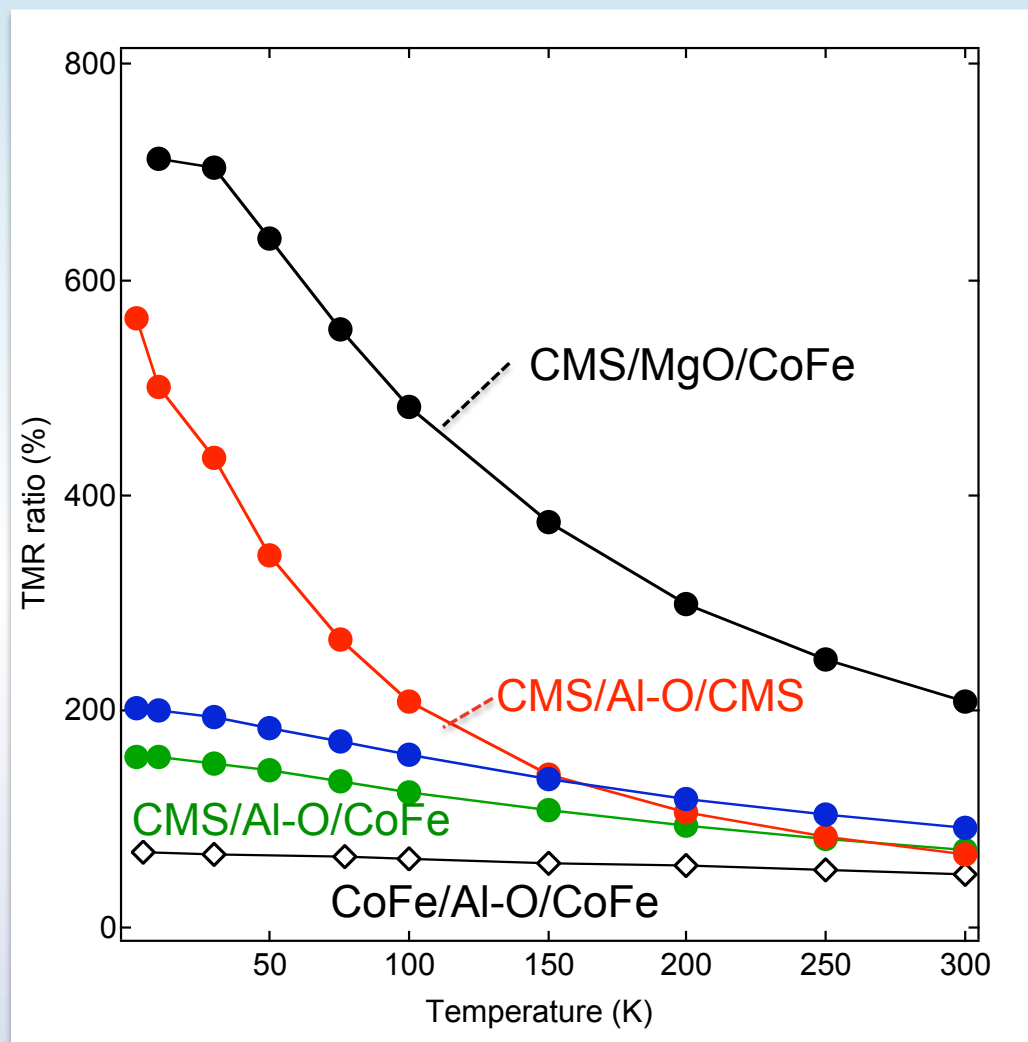
#### $\text{Co}_2\text{MnSi}/\text{MgO}/\text{Co}_2\text{MnSi}$ -MTJ

TMR = 182%(RT), 705%(4.2K)

Ishikawai *et al.* *JAP* (2009)

#### cf. $\text{CoFeB}/\text{MgO}/\text{CoFeB}$ -MTJ

TMR = 604%(RT), 1144%(4.2K)



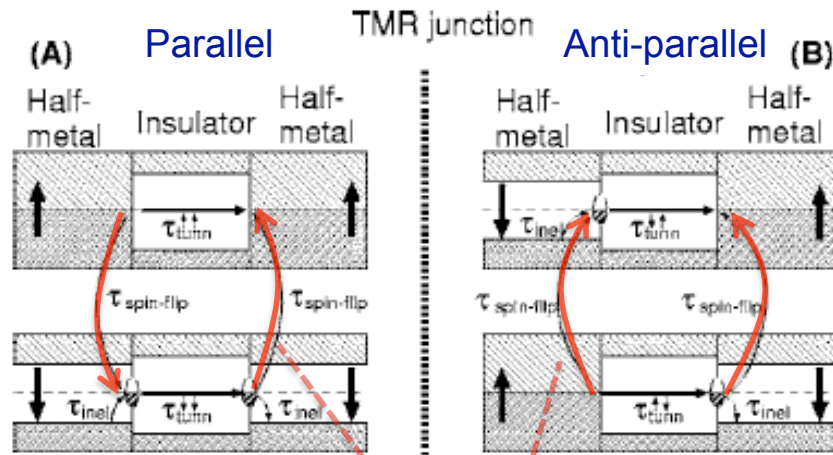
**Disappearance of half-metallicity at RT is the largest problem in CMS-based MTJs.**

# Origin for disappearance of half-metallicity at RT

PHYSICAL REVIEW B 72, 174428 (2005)

## Half-metallic ferromagnets for magnetic tunnel junctions by *ab initio* calculations

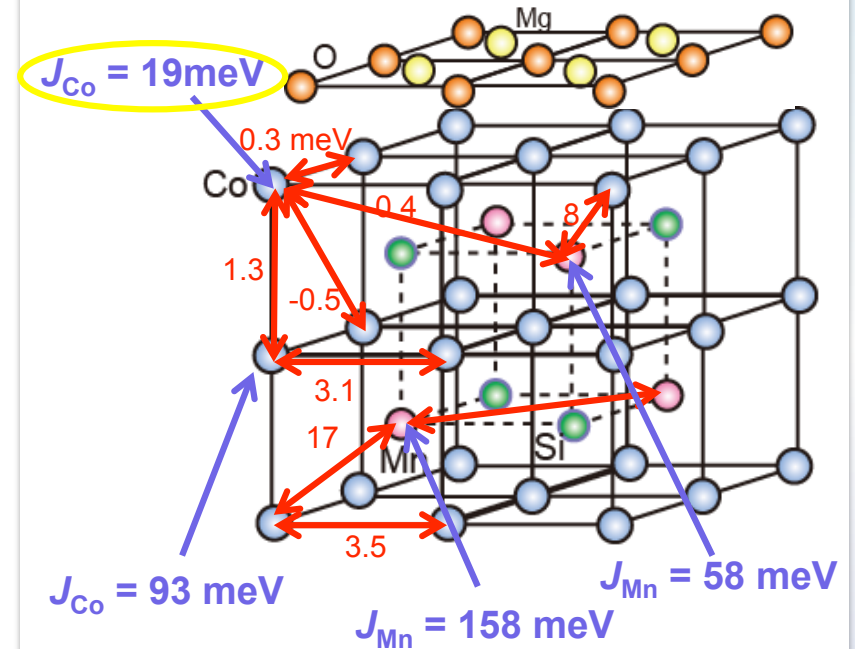
Phivos Mavropoulos,\* Marjana Ležaić, and Stefan Blügel  
 Institut für Festkörperforschung, Forschungszentrum Jülich, D-52425 Jülich, Germany  
 (Received 23 August 2005; published 29 November 2005)



Spin-flip at interface due to magnon excitation.

Sakuma et al. J. Appl. Phys. 105, 07C910 (2009)

## Co<sub>2</sub>MnSi/MgO



Weak exchange energy of Co at the interface.

Large magnon excitation by thermal energy is the most possible origin for disappearance of half-metallicity at RT in CMS-based MTJs.

For developing Spin-RAM

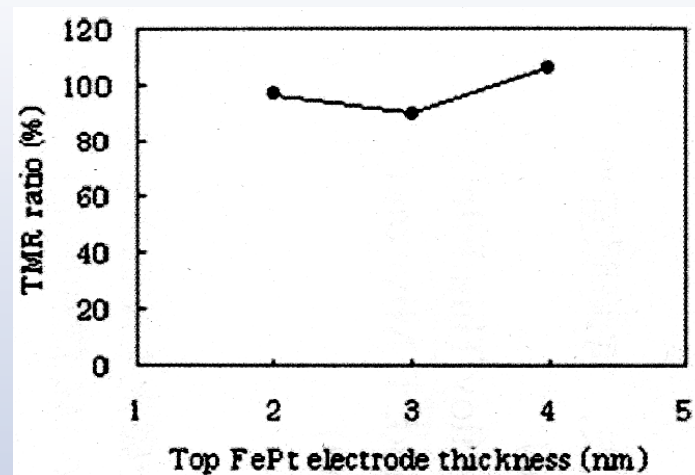
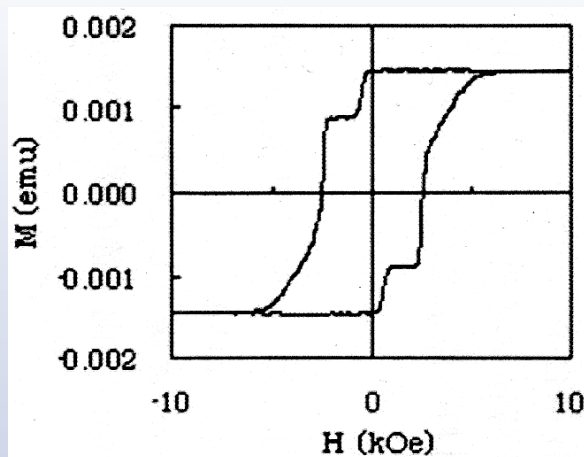
MTJs with perpendicular magnetized electrodes are recent trend.

- High thermal stability :  $K_u V/k_B T > 60$
- Small critical current density of spin-transfer torque switching :  $J_{C0} < 10^6 \text{ A/cm}^2$

### FePt/Fe/MgO/FePt – MTJ

(MMM 2008, TOSHIBA)

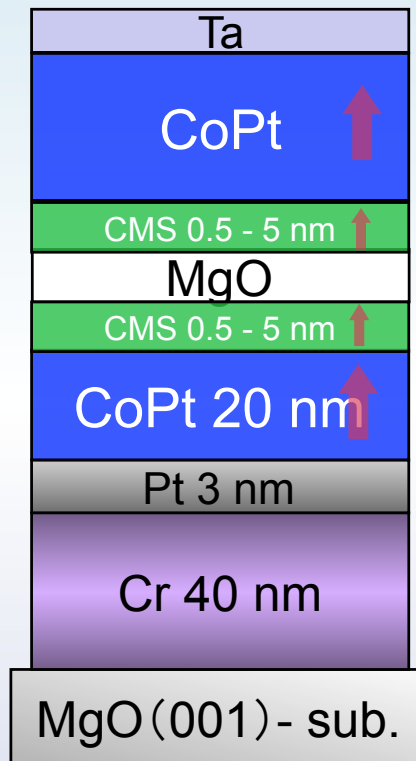
→ TMR ratio > 100% at R. T.



## Experiment 2 MTJs using L1<sub>0</sub>/L2<sub>1</sub>-ordered alloy hybrid electrodes



Lattice misfit between CMS(001) and CoPt(001) ~ 4%  
(001)-CoPt/CMS/MgO/CMS/CoPt fully-epitaxial MTJ can be fabricated.



★ High thermal stability due to large magnetic anisotropy

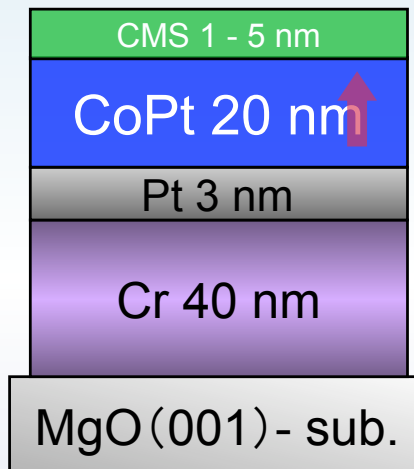
Suppression of magnon excitation at CMS/barrier interface.

★ Large spin-polarization (TMR ratio) at RT

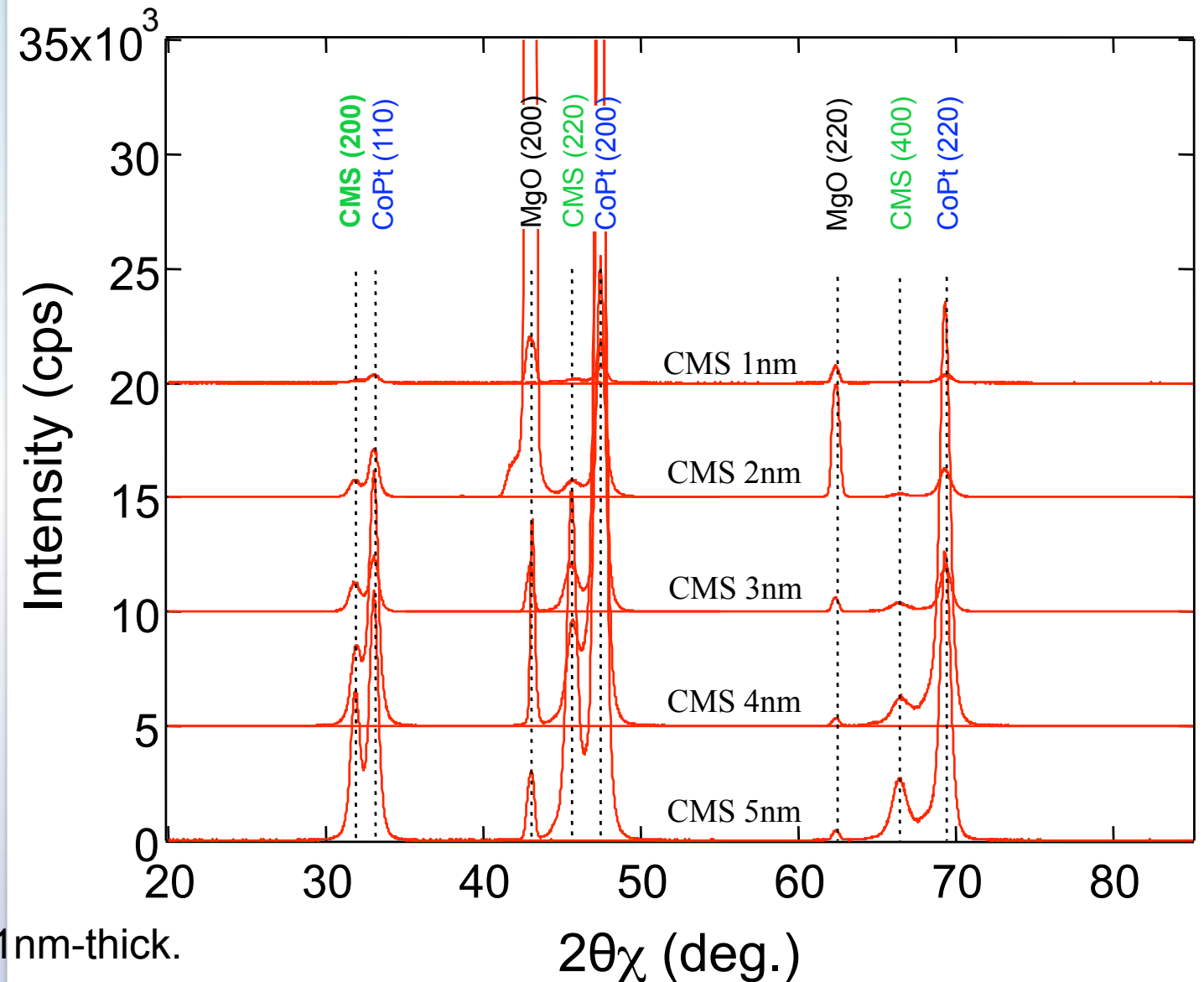
# Experiment 2 MTJs using L<sub>1</sub><sub>0</sub>/L<sub>2</sub><sub>1</sub>-ordered alloy hybrid electrodes

Structural analysis : XRD in-plane geometry (X-ray incident angle  $2\theta < 0.4\text{deg.}$ )

CMS: Deposited@RT  
→annealed@450°C



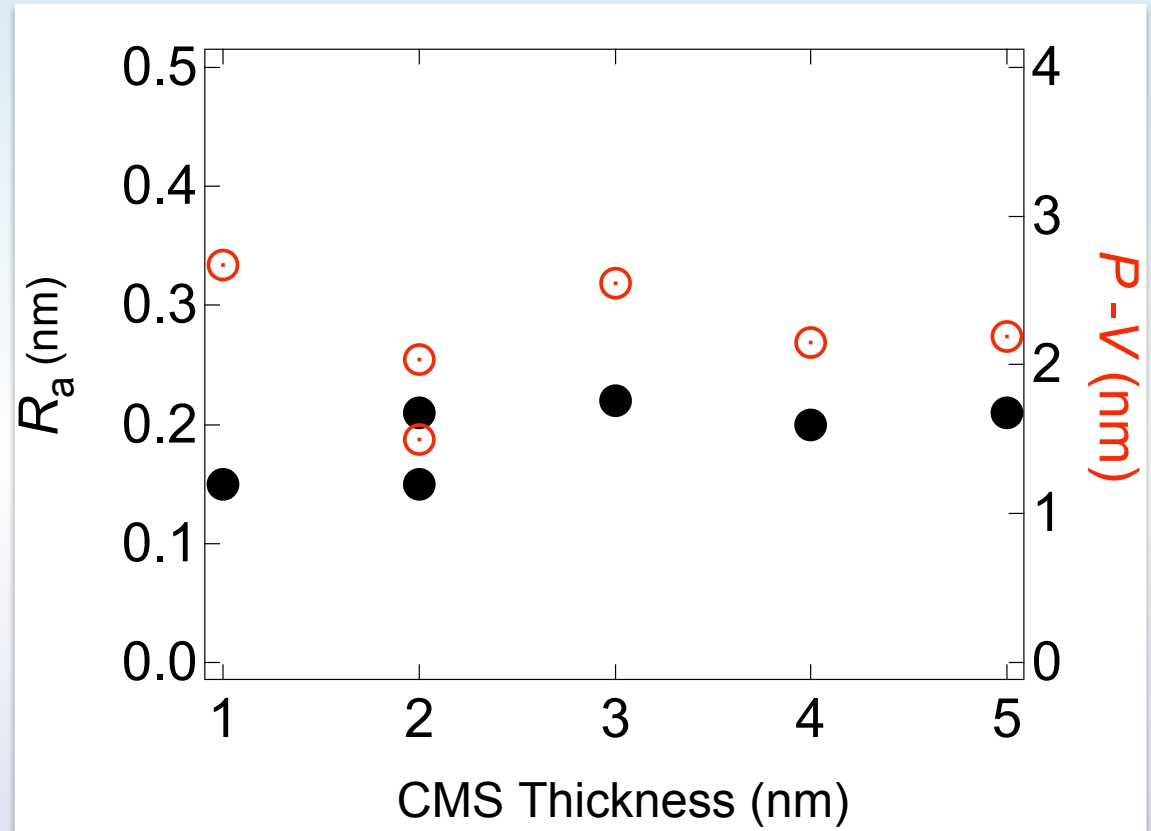
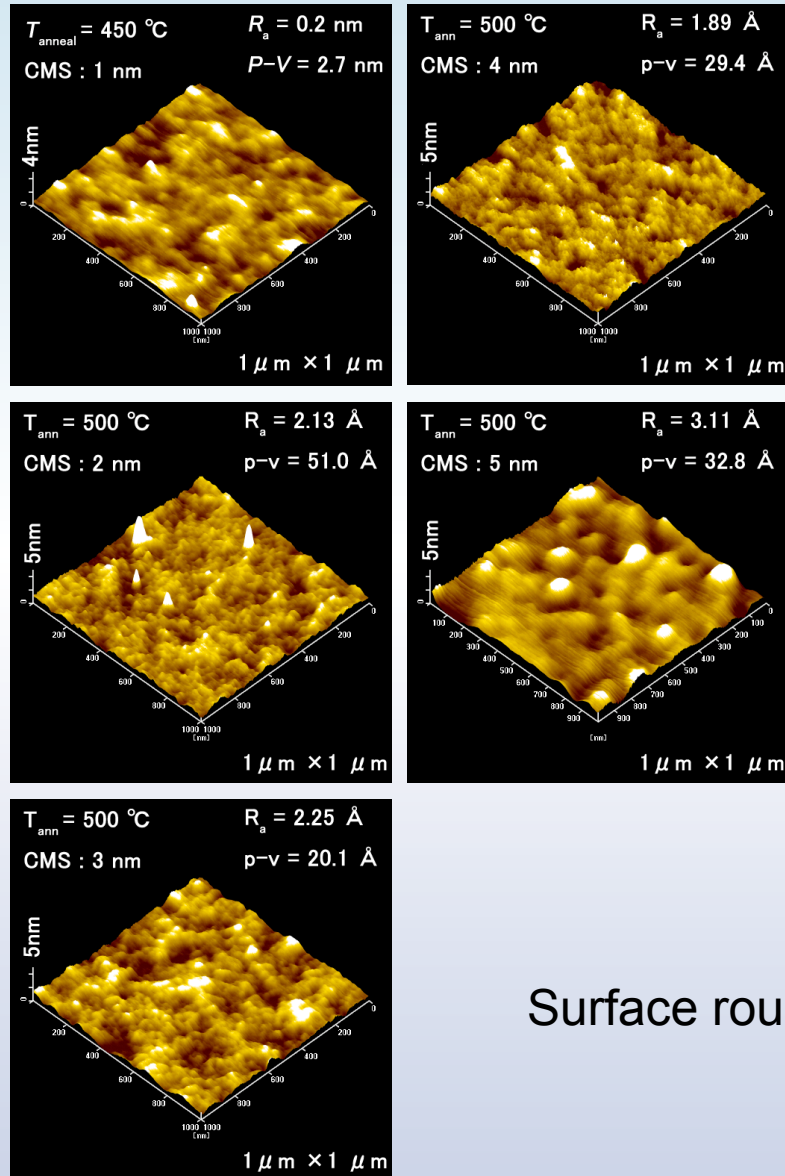
- fully-epitaxial growth
- B2-ordering even in 1nm-thick.





# Experiment 2 MTJs using L1<sub>0</sub>/L2<sub>1</sub>-ordered alloy hybrid electrodes

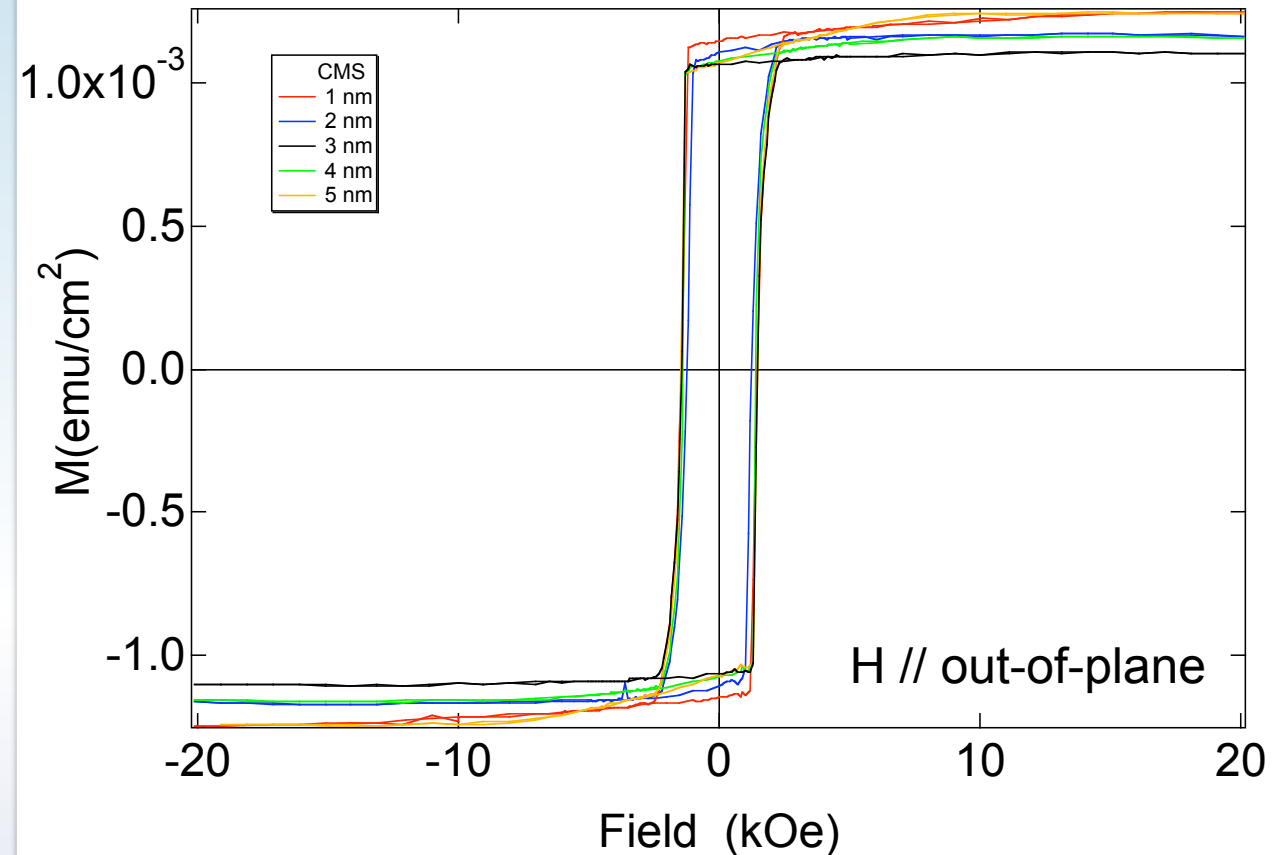
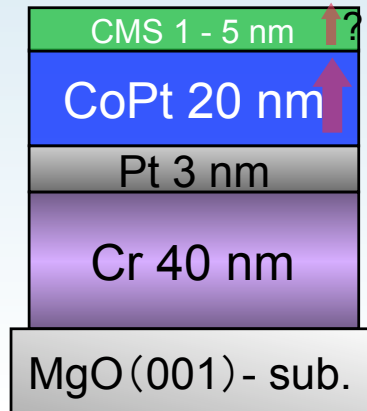
## Surface flatness : AFM



Surface roughness is enough small to deposit MgO barrier.

## Experiment 2 MTJs using $L1_0/L2_1$ -ordered alloy hybrid electrodes

$M$ - $H$  curves : SQUID, VSM



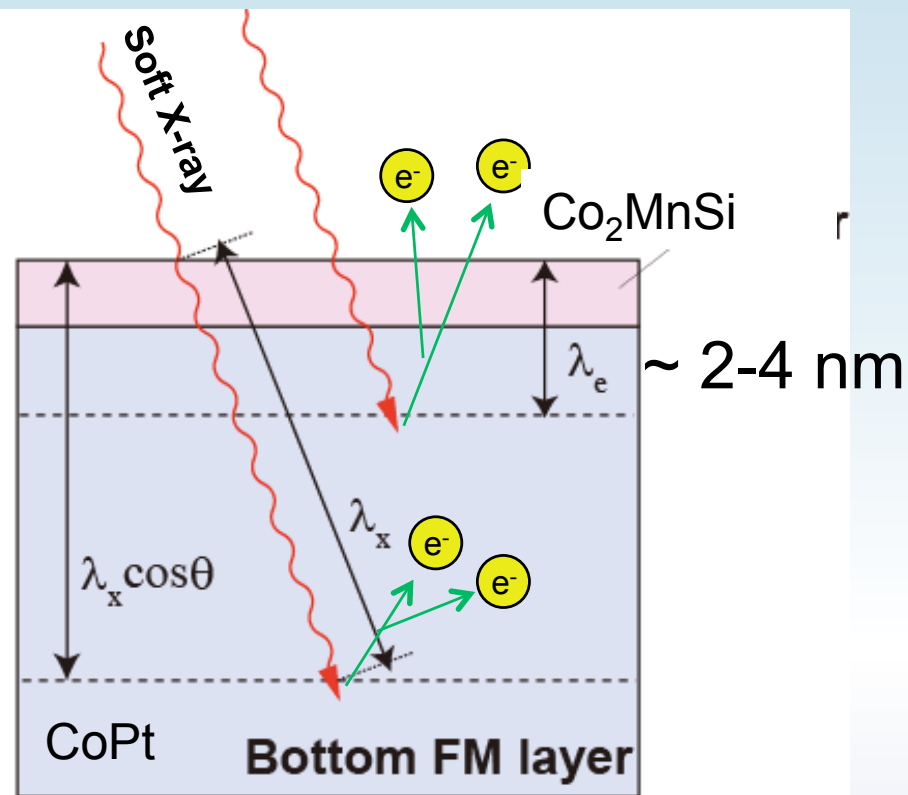
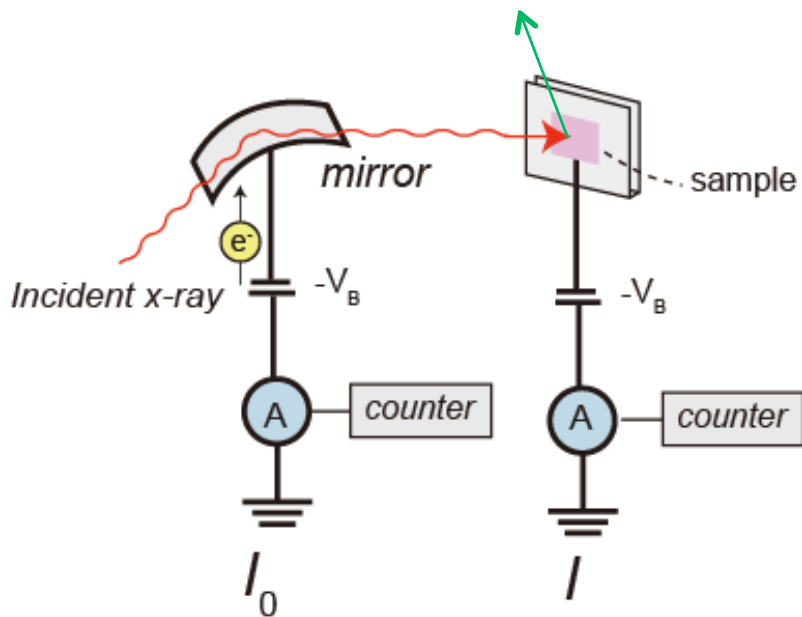
Magnetic behavior of CMS layer can not be identified because of...

- (1) too small  $M_{\text{CMS}}$  compared with  $M_{\text{CoPt}}$
- (2) the error in subtraction procedure of MgO-substrates contribution

Other experimental techniques to observe surface moment is needed.

***XMCD***

Total electron yield (TEY) method



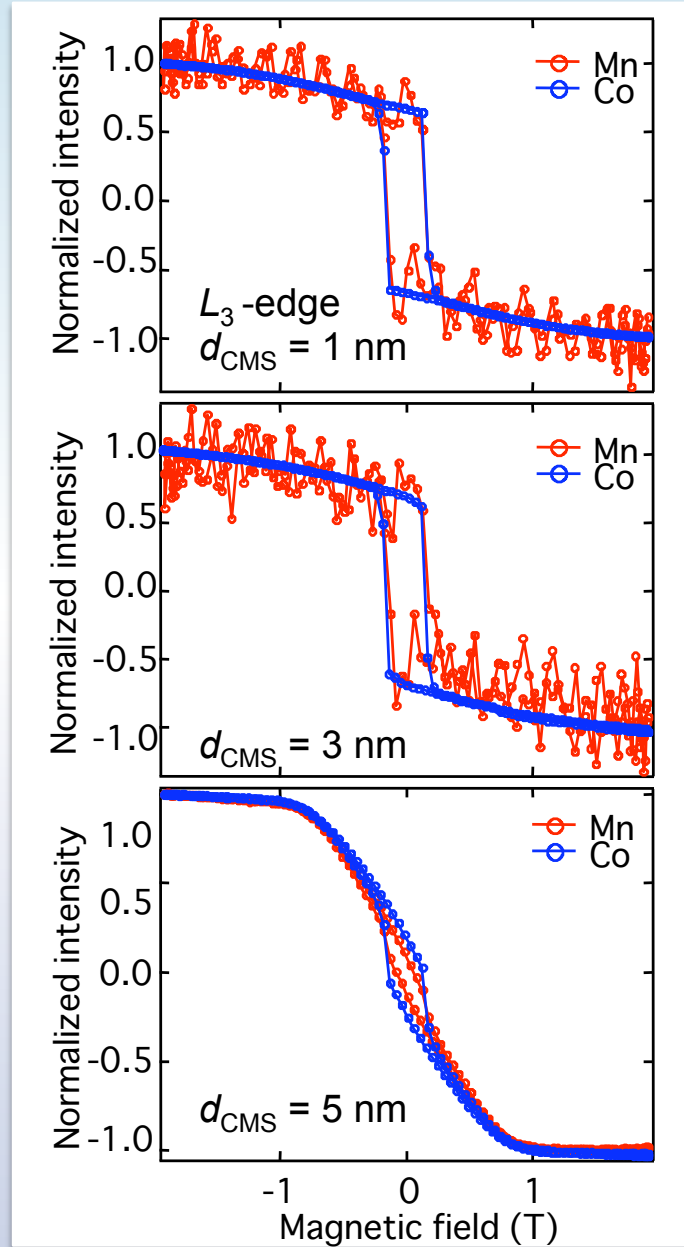
**XMCD : SPring-8 BL25SU**

- $\lambda_x$  : penetration length of X-ray
- $\lambda_x \cos\theta$  : penetration depth of X-ray
- $\lambda_e$  : escape depth of photo electron

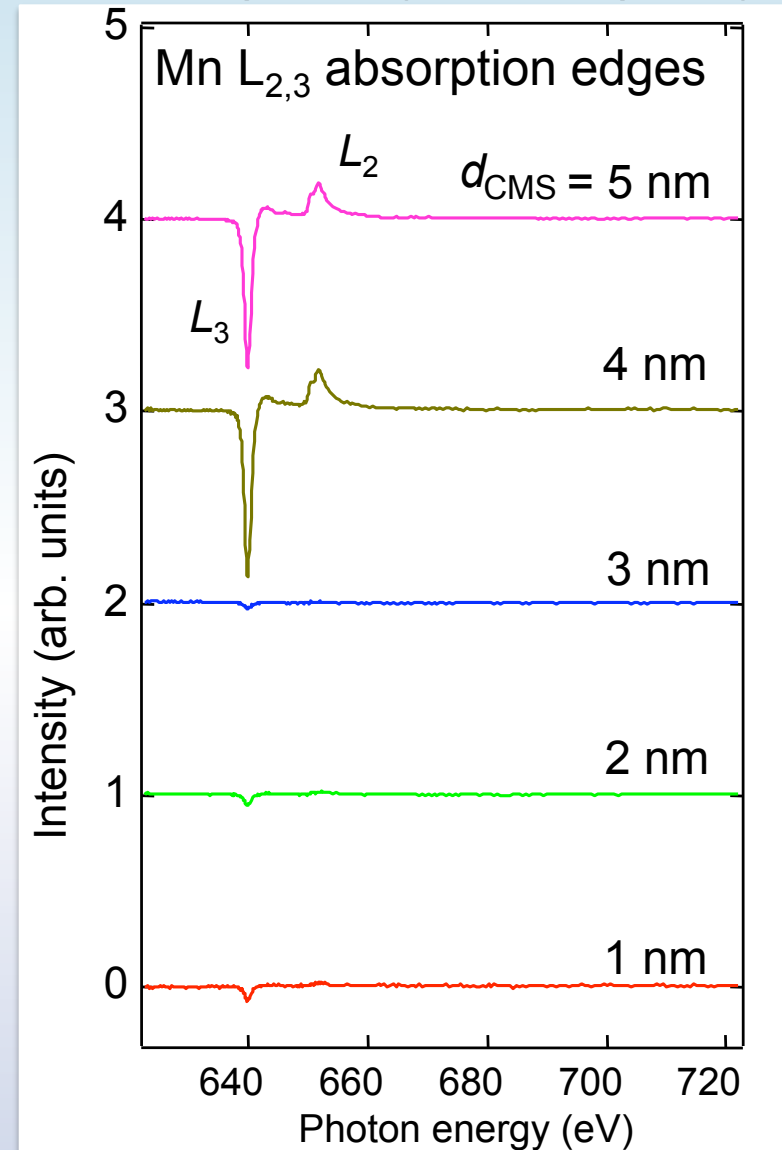
*Elements specific (Co and Mn) M-H curves can be measured.*

# Experiment 2 MTJs using L1<sub>0</sub>/L2<sub>1</sub>-ordered alloy hybrid electrodes

Element specific M-H (H//out-of-plane)



XMCD spectra (H//out-of-plane)



•  $d_{\text{CMS}} < 3 \text{ nm}$  : CMS has perpendicular magnetic anisotropy, but magnetic moment is very small.

## What is the next generation reading head?

**Requirements** : Large MR ratio for high S/N ratio, Small RA for high speed reading

### ● MgO-MTJs

Large MR ratio, but difficult to achieve small RA.

CoFeB/MgO/CoFeB : RA =  $0.4 \Omega \cdot \mu\text{m}^2$ , MR = 50%  
*Nagamine et al. APL(2006)*

RA =  $2.1 \Omega \cdot \mu\text{m}^2$ , MR = 206%  
*Isogami et al. MSJ conf.(2008)*

### ● CPP-GMR

Small RA, but difficult to achieve large MR ratio.

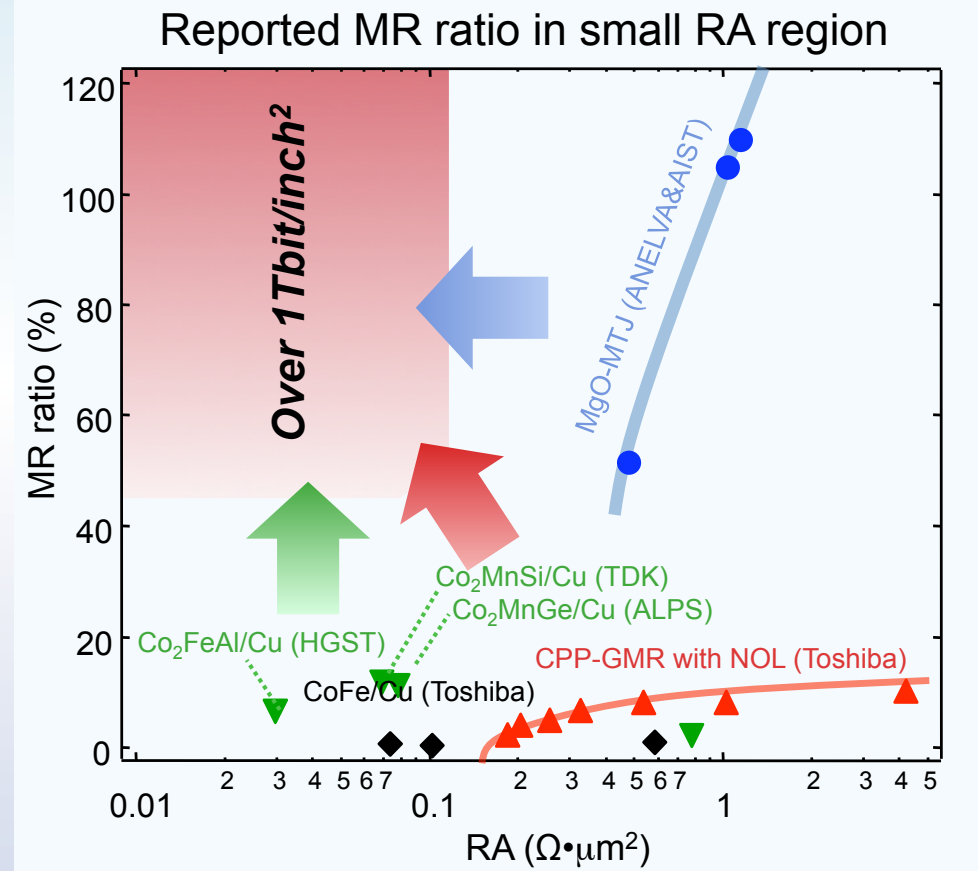
CoFe/Cu/CoFe : RA =  $0.12 \Omega \cdot \mu\text{m}^2$ , MR = 1.3%  
*Yuasa et al. JAP(2002)*

#### (i) Using Nano-oxide layer(NOL)

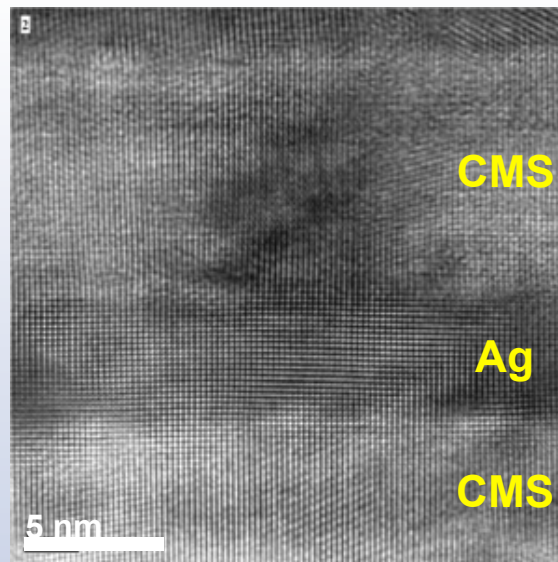
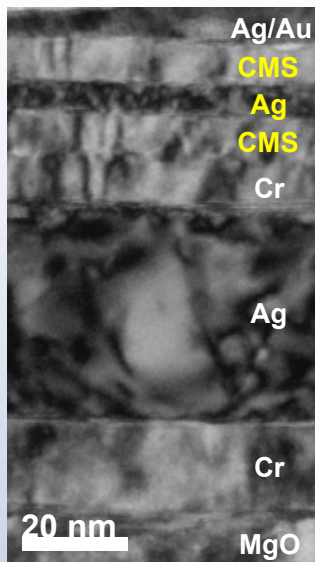
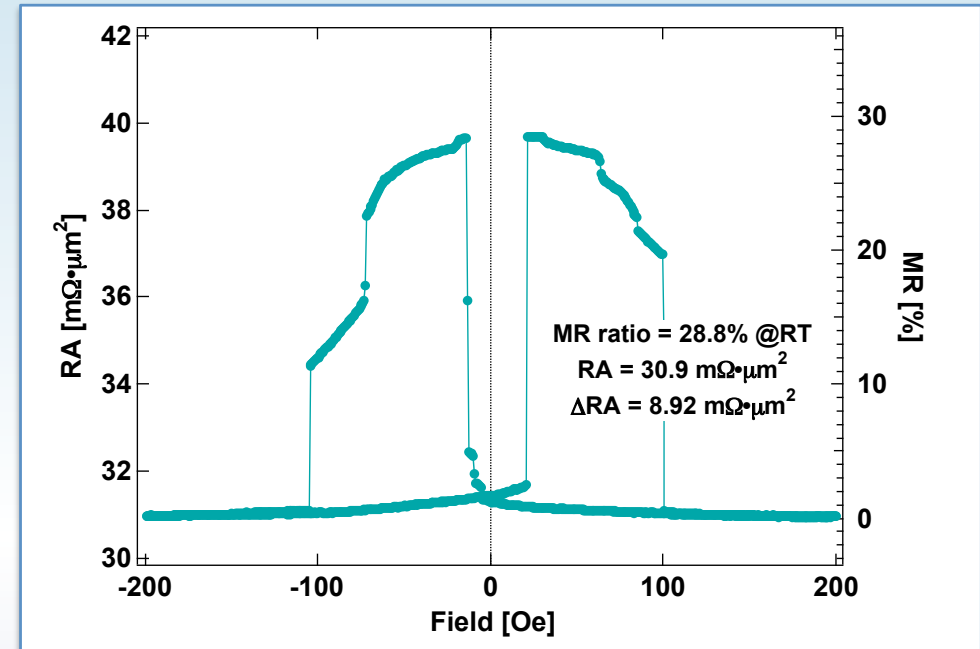
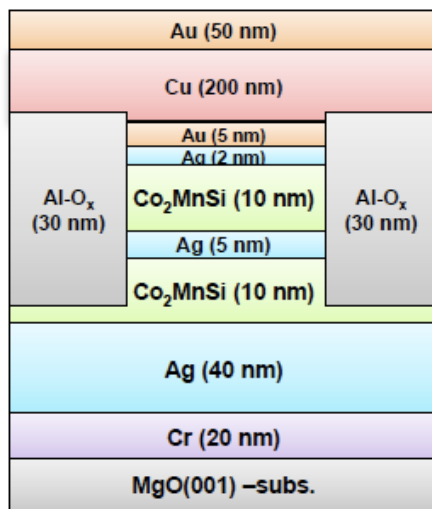
CoFe/NOL/CoFe : RA =  $0.58 \Omega \cdot \mu\text{m}^2$ , MR = 8.2%  
*Fukuzawa et al. JPD(2007)*

#### (ii) Using half-metallic full-Heusler alloy

Co<sub>2</sub>MnGe/Cu/Co<sub>2</sub>MnGe : RA =  $0.079 \Omega \cdot \mu\text{m}^2$ , MR = 11.5%  
*Saito et al. Intermag(2004)*



## CMS/Ag/CMS fully-epitaxial CPP-GMR device



HR-TEM image

◆ The highest MR ratio (28.8%@RT) is the best record in all of the reports to date.

cf. CoFe/Cu/CoFe : MR ~ 2.3%@RT

- XAS based on TEY is an useful technique to observe interfacial chemical state between barrier and FM electrodes in MTJs.
- Element specific magnetization measurement based on XMCD is an helpful technique to fabricate new types of half-metals such as, perpendicular magnetized half-metals and ferri-magnetic half-metals.